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Additive-free electroless deposition on graphene/copper foil: Photo-induced and defect-assisted approach for environmentally friendly plating

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ABSTRACT

The environmental toxicity of chemicals used in electroless plating process is often overlooked. In fact, the wastewater from the electroless plating process contains metal complexes, reducing agents and stabilizers that can create hazard to our environment. To address this concern, an additive-free electroless deposition process on graphene/Cu foil is proposed. The defect structure of graphene plays a pivotal role by acting as a reservoir for concentrating photoelectrons generated from ultra-thin cuprous oxide (Cu₂O) located at the interface of graphene and Cu foil upon exposure to light irradiation. In the Cu ion solution, Cu ions were anchored on the defect sites and formed a Cu-C₄ structure with graphene. The Cu-C₄ sites accumulate abundant photoelectrons and trigger the reduction of Cu ions to form island-like growth of Cu, which immediately was oxidized to Cu₂O, leaving only a small portion of metallic Cu. Concurrently, the presence of Cu ions induces the generation of additional defect sites on graphene, providing further growth sites for Cu deposition. This process continues until the entire graphene surface is covered by the deposited Cu₂O. Ultimately, the deposition of metallic oxides with lower redox potential can be achieved. The proposed deposition mechanism assisted by irradiation of photocatalyst and defect structure of graphene material provides a truly environmental hazards associated with electroless plating, this method holds great promise for sustainable materials manufacturing.

1. Introduction

Pollutants such as sludge waste baths and wastewater from the electroplating industry are invariably hazardous [1]. On the contrary, electroless plating process, also known as autocatalytic plating, which utilizes its self-catalysis property to induce a controllable redox reaction without applying electric power is considered environmentally friendly [2,3]. The minimum composition of an electroless plating bath consists of metal ions and reducing agents at a specific pH value. However, additives such as stabilizers, complexing agents and buffers are commonly added to electroless plating bath, resulting in a large amount of metal-organic compounds and chemical reagents in wastewater [4],

which increases the cost of chemicals, wastewater management and environmental pollution risks. Therefore, reducing the amount and type of chemical reagents used is a prerequisite for the development of a truly green electroless plating process.

Recently, the approach of substrate-enhanced electroless deposition has attracted much attention because it does not involve reducing agents [5–7]. Meanwhile, the defect structures of carbon materials could provide anchor sites for metal atom via the strong charge transfer by 2π antibonding state of the carbon atoms for stabilizing metal atom [8]. These obtained single-atom metal atom/carbon materials exhibit high activity in catalytic reactions [6,9]. Additionally, the formation of metallic clusters and nanoparticles could be obtained for some noble

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Received 8 August 2023; Received in revised form 30 November 2023; Accepted 14 December 2023 Available online 17 December 2023 2213-3437/© 2023 Elsevier Ltd. All rights reserved. metals due to their higher redox potential [10,11]. Also, the defect structures possess lower work function than defect-free carbon structure, allowing noble metals to be preferentially deposited on the defect structure. However, the distribution of deposited metallic clusters and nanoparticles on the surface of carbon materials is dominated by the degree of defect structure. It means that the electroless deposition of uniform metallic layer cannot be achieved, especially for metal species with lower redox potentials, limiting its practical applications.

Recently, the deposition of cuprous oxide (Cu₂O) with lower redox potential on carbon materials has been of intense interest in photothermal conversion, antibacterial protection and photo/electrochemical catalytic reaction [12-16]. In particularly, oxide-derived Cu is recognized as the most important catalysts toward electrochemical CO₂ reduction reaction since only Cu-based catalysts can effectively convert CO₂ into valuable hydrocarbon and alcohol products [17-20]. Meanwhile, carbon materials with large specific surface area and high conductivity endow synergistic activity with catalysts. The common method for coating the prepared Cu₂O onto carbon-based materials involves spraving a Cu₂O suspension. However, due to the weak interaction between Cu₂O and the substrate, it is prone to detachment, especially during electrochemical gas evolution reactions. To enhance the adhesion between Cu₂O particles and between Cu₂O and the substrate, a small amount of polymer, such as Nafion, is typically added to the Cu₂O solution [21,22]. Even more concerning is the use of various chemical reagents during the synthesis of Cu₂O nanoparticles. These reagents include stabilizers (polyethylene glycol or polyvinylpyrrolidone), reducing agents (ascorbic acid, sodium borohydride, polyvinylpyrrolidone or glucose), pH adjusters (sodium hydroxide or potassium hydroxide) and solvents (diethylene glycol or ethylene glycol) [23–29]. The use of these chemicals not only increases production and subsequent treatment costs but also poses direct and potential risks to the environment when handled improperly.

In this work, an additive-free electroless deposition of Cu_2O on commercial graphene/Cu foil with the interface of Cu_2O is proposed by a photo-assisted process. We leverage on the photoelectric properties of Cu_2O to provide photoexcited electrons upon light irradiation, which are accumulated at the defect sites of graphene. The electron-rich sites serve to reduce Cu ions as well as to avoid the re-dissolution of deposited Cu_2O [30,31], in which the growth site started from the Cu-C bond at the defect structure. Simultaneously, the Cu ions in $CuSO_4$ solution induced the formation of defect structure as growth sites, which extend the deposition reaction to the entire graphene surface and achieve high-density deposition. Based on the evidences of in situ Raman spectroscopy, ex situ X-ray photoelectron spectroscopy (XPS) and X-ray absorption spectroscopy (XAS), the mechanism of electroless process on graphene/Cu foil surface is clarified.

2. Results and discussion

Fig. 1a showed the photo image of graphene/Cu foil obtained from Graphenea Inc. The growth method of graphene on Cu foil was adopted via chemical vapor deposition. The Raman spectrum shows two main peaks at 1589 and 2644 cm⁻¹, which are the assigned G band and 2D band (Fig. 1b). The number of graphene layer could be derived from the ratio of 2D to G band intensity. The value of I_{2D}/I_G is around 2, meaning the graphene on Cu foil is single layer graphene. A relatively weak D band from disordered graphite appearing at 1324 cm⁻¹, indicating a slight defect structure in graphene.



Fig. 1. (a) photo image of graphene/Cu foil. (b) Raman spectrum of graphene. (c) Scheme of Cu foil oxidation by the oxygen passing through the defect structure of single layer graphene. (d) The characteristic of the interface of graphene and Cu foil by TERS.

Literatures reported that the few-layer graphene on Cu surface would block the penetration channel of oxygen and water vapor reaching Cu surface, thus exhibiting the function of anticorrosion [32]. However, single-layer graphene structure containing a small number of defect sites can become a channel for oxygen to pass through, thereby oxidizing the surface of the Cu foil (Fig. 1c) [32]. It is worth noting that the small amount of defect structure presented in single-layer graphene that causes slight oxidation of Cu will be undetectable by conventional Raman spectroscopy without auxiliary enhancement methodology. By introducing the technique of tip-enhanced Raman spectroscopy (TERS) (Fig. 1d), some characteristic peaks of Cu₂O were observed at 211, 428, 523 and 621 cm⁻¹ [33], indicating that a very thin Cu₂O film was indeed formed at the interface of graphene and Cu foil.

To clarify the structural state of graphene on Cu_2O surface, atomistic scale modelling and density functional theory calculations were employed to investigate the penetration of oxygen atoms from the slightly oxidized Cu layer through the defect channels of single layer graphene as shown in Fig. 1c. We modelled a system of a pristine copper block and layered it with 8 Cu₂O molecules. However, the interactions or bond formations was not pre-determined as shown in Fig. 2a and 2b. The combination of pristine Cu+Cu₂O molecules were geometrically optimized to attain a lower energy state using Vienna Ab-initio Simulation Package (VASP). After the geometry optimization, it is found that the Cu₂O molecules readily bonded with the pristine Cu system which resembles to a slightly oxidized Cu foil as mentioned earlier (Fig. 2b).

Subsequently, addition of a single layer of defective graphene is done and again the system is geometrically optimized. It is found that the presence of a single layer of defective graphene above the slightly oxidized Cu system upset the stability of the Cu system (Fig. 2c) in the following. Firstly, the oxygen atoms from the Cu₂O molecules slightly permeates through the defect channels of the graphene layer as shown in the green circles in Fig. 2d. Secondly, some Cu atoms dissociate the Cu-Cu bonding and form unstable bonds with the defect sites in the graphene layer as shown in the blue circles in the same Figure. As these reactions of the Cu-Cu₂O-graphene system occurs in a stable energy state (i.e. the structure is geometrically optimized for stability), they can potentially oxidize further in presence of suitable external energy.

The co-existence of defect structures of graphene and Cu₂O on the Cu foil surface provided a potential for electroless Cu plating on graphene. The defect structures act as reducing agents due to their lower work function and anchoring points for in situ growing metal cluster [10,34]. As investigated by density functional theory calculations, the binding energy of metallic atom to the defect site of graphene was higher than that of metallic atom-metallic atom and metallic atom-C (without defect structure) [7,35]. For instance, the Pt, Pd, Sn and Au atoms exhibit the binding energies of 7.234 eV, 5.237 eV, 4.487 eV and 2.345 eV to defect sites of graphene, which are higher than those binding energies of 1.552 eV, 1.077 eV, 0.200 eV and 0.075 eV to graphene without defect



Fig. 2. (a) Cu and Cu_2O system modelling (b) Geometrically optimized Cu and Cu_2O system with single layer defective graphene (c) Complete geometrically optimized Cu and Cu_2O system with a single layer defective graphene (d) Green circle show oxygen atoms permeating through defect sites of graphene, blue circle show the bond formation between Cu and defect sites in the graphene and yellow circles show bond formation between O and defect sites in the graphene.

sites. Consequently, the metal clusters would generate with the additional metallic atoms received by the C sp²-orbitals surrounding the defect sites. So far, only the Pt, Au and Pd clusters anchoring on the defect site were observed due to their higher redox potentials which were 1.19 V, 1.50 V and 0.92 V, respectively [29]. Additionally, the entire substrate was not completely covered by reduced metal because the presence of the reduction sites limited the extensive growth of metallic clusters [10]. However, electroless Cu plating on graphene via the above process was not achieve due to the relative lower redox potential at 0.34 V [36].

Now the existence of Cu_2O at the interface of graphene and Cu foil as observed could provide light-induced photoelectrons which can be further captured and concentrated at the defect sites. This phenomenon would enhance the reducing capacity and be able to assist the nonspontaneous reduction of Cu ion on the graphene surface. For the graphene/Cu foil, the interface of Cu_2O shows a wide absorption range in UV-Vis spectrum from 400 to 600 nm (Fig. 3a), which is close to that of Cu_2O nanoparticles and electrochemically deposited Cu_2O thin films [37,38]. This observation suggests the presence of Cu_2O at the interface between graphene and Cu, which had been confirmed by TERS (Fig. 1d). Besides, it also implies that photoelectrons can be induced when exposed to light of appropriate wavelength.

To study the feasibility of the above-mentioned, namely electroless Cu plating on graphene via light excitation, LEDs with various wavelength of emission light are used to generate different levels of photo-excited electrons. The PL spectra of the emission sources exhibit blue, green, red and white with the maximum wavelength of 463, 524, 632 and 452 coupling 547 nm, respectively. The experimental setups are shown in Fig. 3b and 3c, in which 1 cm² of graphene/Cu foils are respectively immersed in 5 mM of CuSO₄ with different irradiation wavelengths under constant stirring at 200 rpm.

Fig. 4 shows the time-dependent microscopic images of graphene/Cu foil surfaces in 5 mM of $CuSO_4$ solution at different emission wavelengths. Comparison of the images from 10 min to 4 h in the dark state show no reduction of Cu ions occurred due to the low reduction ability of

graphene defect structure without any assistance, and this could be the reason of no experimental evidence of spontaneous reduction of Cu on defect structure. Under the auxiliary irradiation of blue, green, red or white light LEDs whose emission wavelength satisfies the absorption of Cu₂O, obvious particles appear on the surface within 1 h, suggesting the generated photoexcited electrons endowed the graphene defect structure with stronger reduction ability. Upon prolonged light irradiation, more particles produced and grown until they were connected to each other and cover the entire surface of the graphene. In fact, some fine particles can be clearly observed after 10 min of green light irradiation, and the particle dispersion density was the highest. Furthermore, the 3D image of optical microscopy showed that the particles are actually formed, instead of being etched on the graphene/Cu foil (Fig. S1). Our results indicated the optimal light source is green light. This is expected because the maximum absorption of Cu₂O film from the interface of graphene/Cu foil was closed to 524 nm as shown in Fig. 3(a). The photo and scanning electron microscope (SEM) images further indicated the reduction and deposition keep happening at the particles edges, resulting in an increase in size over time (Fig. S2).

To identify the elements of the formed particles, EDS mapping is adopted to probe the element distribution. The elements of the generated particles include the Cu element as well as the oxygen element, suggesting the particles could be in oxidized state (Fig. S3). The in situ Raman spectroscopy with excitation wavelength of 532 nm shows the formation of Cu₂O with the characteristic peaks at 523 and 621 cm⁻¹ in less than 5 mins (Fig. 5). Also, the intensity increased with immersion time, meaning that increasing amount of Cu₂O is produced. Besides, the XRD pattern shows the diffraction peaks at 29.6°, 36.7°, 42.6° and 61.7°, which were assigned to Cu₂O (JCPDS NO. 78–2076), revealing that the structure of the electroless Cu is Cu₂O, naming EL-Cu₂O (Fig. 5b).

Notably, the deposited Cu_2O could further provide the photoelectrons under exciting irradiation when the initial Cu_2O layer under the graphene is covered by the deposited particles, and thus assist the continuous growth of deposition over the surface of the samples (Fig. S4). This phenomenon was consistent to the observations from the



Fig. 3. (a) The absorption spectrum of graphene/Cu foil (black line) and the wavelength of emission of blue (blue line), green (green line), red (red line) and white (gray line) light LEDs. (b) Schematic and (c) photo images of the experimental setups for electroless Cu plating.



Fig. 4. Microscopy images (100 μ m imes 100 μ m) of graphene/Cu foil surface in CuSO₄ solution at different time under various wavelength of emission.



Fig. 5. (a) In situ Raman spectra of graphene/Cu foil in the CuSO₄ solution. (b) XRD pattern of the deposited particles.

optical and SEM images.

To clarify whether Cu₂O came from the corrosion of Cu foil or the reduction of Cu ions, the ICP-MS is employed to monitor the real-time concentration of Cu ion in CuSO₄ solution during electroless process. Meanwhile, the concentration of Cu ion could be calculated from the calibration curve (Fig. S5). It is found that the concentration of Cu ion in bath decreases with reaction time, demonstrating that the generated Cu₂O is attributed to the reduction of Cu ion (Fig. 6a). Besides, the amount of Cu ion is rapidly consumed within 6 h after the reaction, and then the amount gradually became constant. The amount of EL-Cu₂O is estimated through the consumed Cu ions, where almost no electroless plating occurred after 6 h, indicating the reaction was close to saturation. Additionally, the cross-section of SEM image via paraffin section shows the island-like Cu₂O generated on the partial surface of graphene within 1 h (Fig. 6b). The amount of island-like Cu₂O increases with the

reaction time (Fig. S6), and the island-like Cu₂O grows further and covers the surface of graphene nearly completely with the thickness of about $1.5 \,\mu$ m after 6 h. Besides, the SEM image illustrated the complete coverage of the graphene/Cu foil surface by electroless Cu₂O after electroless for 6 h (Fig. S7). At this point, the formed Cu₂O layer hinders the Cu ion to anchor on the defect sites of graphene, resulting in the stagnation of the electroless process which is consistent to the result of real-time ICP-MS. The EDS mapping of the cross-section of the growth sample clearly distinguish the deposited Cu₂O layer and metallic Cu foil (Fig. 6c).

To unravel the evolution process and growth site of Cu, the chemical state and local structure of EL-Cu₂O are probed by XAS along with Cu foil, Cu₂O and CuO reference spectra (Fig. 7a). According to the calibration standard obtaining from the maximum value of the first derivative of the X-ray absorption near-edge structure (XANES) of Cu foil (0),



Fig. 6. (a) The concentration of remained Cu ion in the solution and the amount of deposited Cu on the sample. (b) The cross-sectional SEM images of graphene/Cu foil after 0, 1 and 6 h of reaction. (c) The cross-sectional EDS mapping of graphene/Cu foil after 6 h of reaction.



Fig. 7. (a) XANES spectra of Cu K-edge, (b) oxidation states of Cu and (c) EXAFS spectra of Cu K-edge. (d) C1 s XPS of graphene/Cu foil after 0, 1 and 6 h of reaction.

Cu₂O (+1) and CuO (+2) (Fig. S8), the oxidation state of EL-Cu₂O was +0.89 (Fig. 7b). This result indicated EL-Cu representing a mixed valence system, involving Cu (1 +) and Cu (0), meaning that the most component of EL-Cu was Cu₂O with a small amount of metallic Cu. Therefore, it could be deduced that Cu²⁺ ions were initially reduced to metallic Cu before undergoing further oxidation in an aqueous solution to form Cu₂O, resulting in only a small amount of metallic Cu remaining. The Fourier-transformed k³-weighted extended X-ray absorption fine structure (EXAFS) spectra of Cu K-edge of the EL-Cu shows a peak at R space of 1.5 Å and a peak at 2.7 Å bond, which were typically regarded as the Cu-O scattering path and secondary coordination shell of Cu-Cu, respectively (Fig. 7c) [39]. This R space is similar to that of Cu₂O standard sample, manifesting the structure of EL-Cu is Cu₂O. The EXAFS of EL-Cu also shows a small shoulder located at 1.1 Å, which is the Cu-C coordination [9].

Additionally, ex situ C 1 s XPS is used to examine the graphene structure in the duration of electroless process. The C 1 s XPS of pristine graphene on Cu foil can be deconvoluted into two peaks at 284.4 eV and 285.2 eV, which are assigned as sp^2 and sp^3 hybridization of carbon

bond (Fig. 7d) [40]. The defect carbon $(sp^3 \text{ carbon bond})$ has already demonstrated by Raman spectrum (Fig. 1b). The new peak generated at 283.8 eV is observed after 1 h of electroless process, and this peak was assigned as Cu-C bond [41]. Evidence for Cu-C bonds in EXAFS and XPS suggested that the interface between EL-Cu and graphene was Cu-C bonds rather than Cu-O-C bonds as reported in the literatures [42,43]. This finding indicated that a strong connection is formed between EL-Cu₂O and graphene. Furthermore, the degrees of Cu-C bond and defect structure increase with the electroless time up to 6 h.

The in situ Raman spectra also reveal that the intensity ratio of D and G bands (I_D/I_G) increases with the electroless time, meaning that the in situ generation of defect structure occurred during electroless process (Fig. 8). In addition, some broad peaks are observed in the region between the G and D bands, and this can be attributed to the formation of amorphous carbon due to the increase of defects or disorders in the honeycomb structure of graphene [44]. It is worth mentioning that the formation of broad band located at 1755 cm⁻¹ as observe during electroless is attributed to the CuC₄ structure [45,46], suggesting the initial growth site of CuC₄. Furthermore, the ratio of the D band to the G band



Fig. 8. (a) In situ Raman spectra of graphene/Cu foil in $CuSO_4$ solution. (b) The function of I_D/I_G with reaction time.

exhibits two distinct behaviors over time (Fig. 8b). The combination of Cu^{2+} ions with the existing defect structures in graphene (formation of Cu-C) rapidly leads to the formation of amorphous carbon in the vicinity of defects. After this stage, most of the original defect structures had transformed into CuC_4 structures. Furthermore, Cu^{2+} ions induce the generation of defect structures, which predominantly governs the deposition rate and coverage completeness. This process occurs more slowly. Meanwhile, Cu^{2+} ions have a tendency to form coordination bonds with carbon atoms in graphene. In this case, the ligands are carbon atoms in graphene. The metal ions can bond to these carbon atoms through the sharing of electron pairs. The formation of coordination bonds between Cu^{2+} ions and carbon atoms can lead to the creation of defects in the graphene structure, including vacancies and disruptions in the regular hexagonal lattice. This phenomenon has also been mentioned in the literature [47].

To evaluate the adhesion between chemically deposited Cu₂O and graphene/Cu foil, the stability of the current values was assessed through electrochemical hydrogen evolution reactions. Fig. S9 displays the *i-t* chronoamperometric response of the Cu₂O nanoparticlesmodified graphene/Cu foil (Cu₂O NPs/graphene/Cu foil) and electroless Cu₂O/graphene/Cu foil (EL-Cu₂O/graphene/Cu foil) at -1.2 V (vs RHE) in a 0.1 M KHCO $_3$ solution with saturated Ar. For Cu₂O NPs/ graphene/Cu foil (Fig. S10), the preparation of suspended Cu₂O NPs followed a procedure in the literature, which involved the use of polyethylene glycol as a stabilizing agent and L-ascorbic acid as a reductant [28]. Additionally, a reaction time of 6 h for the electroless deposition of Cu₂O was chosen to ensure complete coverage of the graphene/Cu foil surface by Cu₂O. Over time, the catalytic current of Cu₂O NPs/graphene/Cu foil without Nafion decreased, suggesting that Cu₂O NPs detachment reduced the active site population during electrochemical hydrogen evolution. In contrast, the electrocatalytic current density of EL-Cu₂O/graphene/Cu foil remains stable, suggesting a strong interaction between either the deposited Cu₂O and Cu₂O or Cu₂O and graphene/Cu foil.

To evaluated the impact of the proposed electroless Cu₂O deposition method on surface roughness using electrochemically active surface

area (ECSA). ECSA is typically directly correlated with the roughness of the electrode surface. In electrochemistry, the surface roughness of an electrode increases the available surface area, leading to a greater number of active sites on the electrode. Therefore, a larger rough surface area generally implies a larger ECSA. The ECSA values for graphene/Cu foil, EL-Cu₂O/graphene/Cu foil and Cu₂O NPs/graphene/Cu foil can be determined by extracting the double-layer capacitance (Cdl) through cyclic voltammetry (CV) measurements in the non-Faradaic potential region. Subsequently, ECSA is calculated using the equation: ECSA = C_{dl}/C_s , where C_s represents the specific capacitance of the electrocatalysts. The value of C_{dl} can be obtained through plotting the average difference between forward and reverse currents at the middle of the potential window against scan rates (Fig. S11). Furthermore, ECSA values can be calculated by dividing Cdl by a specific capacitance of 40 $\mu F\ cm^{-2},$ typical for transition-metal oxides. Therefore, at a fixed geometric area of 1 cm⁻², the ECSA values for graphene/Cu foil, EL-Cu₂O/ graphene/Cu foil and Cu₂O NPs/graphene/Cu foil are 20.0, 52.5, and 70.0 cm², respectively. Additionally, the roughness factor, which represents the ratio of the real surface area to the geometric surface area, can be determined by dividing the ECSA by the geometric electrode area (1 cm⁻²). The roughness factors for graphene/Cu foil, EL-Cu₂O/graphene/Cu foil and Cu₂O NPs/graphene/Cu foil are 20.0, 52.5 and 70.0, respectively. These results indicate that the surface of graphene/Cu foil exhibited some degree of roughness, as observed in surface morphology of SEM (Fig. S2a). In comparison, the EL-Cu₂O/graphene/Cu foil results in a lower surface roughness compared to Cu₂O NPs/graphene/Cu foil.

The growth sites of Cu and the mechanism of electroless process is proposed based on aforementioned results. As shown in Fig. 9, the defect site of graphene on slightly oxidized Cu foil provided an anchor site for Cu ion via Cu-C bonding to stabilize Cu ion. The anchored Cu of CuC₄ was one type of Cu single-atom structure, which was considered as the electron reservoir and highly active. Under irradiation, the photoelectrons generated from the thin Cu₂O layer at the interface of graphene and Cu foil were accumulated on the anchored Cu site. This electron-rich site then reduced the Cu ions to metallic Cu particles. In addition to the defect structure, the adsorption energy of Cu to the sp² carbon structure



Fig. 9. Schematic of the electroless process of Cu₂O on graphene/Cu₂O/Cu foil. (the red circles represent the formation of CuC₄ structure through trapping Cu ions on defect structure of graphene.).

was lower than the cohesive energy of Cu species, resulting in the islandlike growth of Cu through Cu-Cu interaction in the initial state [47–49]. Subsequently, most of the reduced Cu was instantaneously oxidized to Cu₂O due to the lack of a strong reducing agent. Therefore, only a small amount of metallic Cu was observed by XAS. At the same time, the interaction of Cu ion to graphene could damage the sp² bonding network of graphene [39], creating more defect sites for further deposition of Cu until the graphene surface was covered by Cu₂O completely. To further demonstrate the practicability of this green electroless process, different metal ion solutions were used on graphene/Cu foil for 6 h under green light irradiation. As shown in Fig. S12, zinc, titanium, cobalt, cerium or cadmium ions can all be deposited on the graphene/Cu foil, respectively. Among them, the deposited cobalt oxide showed the lowest dispersion density. This phenomenon can be attributed to the non-photoelectric property of cobalt oxide which could not provide the extra photoexcited electrons to promote epitaxial growth. In contrast,



Fig. 10. CO₂RR activity of EL-Cu₂O in 0.1 M KHCO₃. (a) LSV curves of EL-Cu₂O/graphene/Cu foil with the scan rate of 10 mV/s. (b) FEs of each product at constant potential.

the photoelectric properties of zinc oxide, titanium oxide, cerium oxide and cadmium oxide lead to high dispersion densities. Surprisingly, the deposition of gold and silver covered the entire surface within 10 min, which was notably faster than the deposition time of other metals. Furthermore, SEM-EDS analysis indicated that no oxides were formed, suggesting that species like metallic Au and Ag, which are not prone to oxidation, remain in their metallic state under this electroless deposition process.

Moreover, to ascertain whether EL-Cu₂O exhibits characteristics suitable for electrochemical CO2RR, we performed linear sweep voltammetry (LSV) on EL-Cu₂O/graphene/Cu foil in two different electrolytes: one saturated with Ar and the other saturated with CO₂, both of which were in a 0.1 M KHCO3 solution were performed. As depicted in Fig. 10a, the electrochemical behavior in the electrolyte saturated with CO₂ displayed distinct differences compared to that saturated with Ar, indicating the activity of Cu₂O/graphene/Cu foil in electrochemical CO₂RR. To investigate the relationship between potential and product formation, experiments on CO2RR were conducted using a customized H-type cell comprising two gastight compartments separated by an anion-exchange membrane. Meanwhile, chronoamperometry was carried out to analyze product distribution at various applied potentials. Fig. 10b illustrated the faradaic efficiency (FE) of the products as a function of the applied potential. In addition to hydrogen evolution, the main products at low reduction potentials were formate and CO. This outcome was attributed to the electroreduction of the of CO2 molecules through the O atom- and C atom-coordinated to Cu, respectively. As the potential increased, the probability of *CO and *CO coupling increases, leading to higher FEs of C₂₊ products (C₂H₅OH and C₂H₄). At higher potentials, CH₄ becomes the predominant product. The product profile confirmed the electrocatalytic behavior of EL-Cu₂O in the CO₂RR.

3. Conclusion

In summary, a graphene-assisted approach of triggering metal deposition under irradiation develops a truly environmentally friendly process. According to the evidences of ex situ XAS, XPS and in situ Raman spectroscopy, the mechanism of electroless deposition is proposed. The in situ generation of graphene defect structure can contribute more growth sites, resulting in deposited metal/metal oxide particles with high dispersion density on graphene surface. Additionally, the formation of Cu-C bond at the interface reinforces the interaction between deposited Cu and graphene layer, avoiding the separation in the working state. The prepared Cu₂O/graphene/Cu structure can be used as oxide-derived Cu electrocatalyst in electrochemical CO_2 reduction reaction.

4. Experimental

4.1. Deposition of Cu on graphene/Cu foil

Monolayer graphene on Cu foil (graphene/Cu foil) was purchased from Graphenea Inc. The graphene/Cu foil with an exposed area of 1 cm^2 was designed for immersion in 5 mM CuSO₄ solution at a stirring rate of 200 rpm. The reaction cell was irradiated by blue, green, red and white LEDs with the wavelength of 463, 524, 632 and 452 coupling 547 nm, respectively. After irradiation with different times, the impregnated graphene/Cu foil samples were washed with DI water and dried in vacuum oven.

4.2. Structural characterization

Field emission scanning electron microscopy (FE-SEM) images were obtained with Hitachi SU8220, and Energy-dispersive X-ray spectroscopy (EDX) mapping was conducted by HORIBA X-max. The crystalline structure analysis was studied by X-ray diffraction (XRD, Bruker D2 Phaser) using Cu K α radiation ($\lambda = 1.54$ Å). The XPS analysis was done

with a PHI 5000 Versa Probe (ULVAC-PHI, Japan) system, using a monochromatic Al K α X-ray. The XAS of Cu K-edge were obtained at SPring-8 (Japan) 12B2 Taiwan beamline of the National Synchrotron Radiation Research Center (NSRRC) operated at 8.0 GeV with a constant current of ~100 mA. In situ Raman spectroscopy was performed by UniDRON (CL Technology) with a 50 × objective lens and a 633 nm laser. XPS spectra were acquired using a PHI 5000 VersaProbe (ULVAC-PHI, Japan) system. During the measurement, a monochromatic Al K α X-ray with a beam diameter of 100 µm was used.

4.3. Atomic scale modelling and simulation methodology

To have a better understanding of the effects of defect sites of single layer graphene, atomic scale modeling is employed. A pure Fm-3 m face centered cubic structure of Cu was used for the calculation where the system consisted of 80 atoms. Above the pure Cu 18 molecules of Cu₂O was layered which represented miniscule oxidations occurring from Oxygen present in the storage before graphene layering. A graphene sheet containing single and double defect sites was introduced over the Cu-Cu₂O system. The density functional theory (DFT) calculations are carried out by projector-augmented wave (PAW) method as implemented in the Vienna Ab-initio Simulation Package provided by Medea (Medea-VASP). The interactions are described using the Generalized Gradient Approximation with Perdew-Burke-Ernzerhof (GGA-PBE) exchange-correlation functional. The atomic orbitals are defined by plane-wave basis sets with cut-off energies of 396.0 eV. Real space projection operators are used and the convergence criterion for selfconsistent field calculations is set to 10^{-5} eV between consecutive steps. The convergence criterion for atomic forces of the systems is set to be less than 0.002 eV A⁻¹. The Brillouin zone is sampled at $4 \times 4 \times 4$ gamma centered k-points where the actual k-point spacing is $0.079 \times 0.063 \times 0.052$ per Angstrom. The structures were optimized after stacking of each layer, while formation of the system to observe the interactions between each molecular species.

CRediT authorship contribution statement

Chen Hsiao-Chien: Conceptualization, Data curation, Methodology, Supervision, Writing – original draft, Writing – review & editing. Shabir Abdul: Methodology, Writing – original draft. Tu Kun-Hua: Methodology, Validation. Tan Cher Ming: Conceptualization, Supervision, Writing – review & editing. Chiu Wei-Hao: Methodology, Validation. Fan Ruei-Cheng: Methodology, Validation. Baruah Nilim Akash: Visualization.

Declaration of Competing Interest

The authors declare the following financial interests/personal relationships which may be considered as potential competing interests: Hsiao-Chien Chen reports financial support was provided by National Science and Technology Council.

Data availability

Data will be made available on request.

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Appendix A. Supporting information

Supplementary data associated with this article can be found in the online version at doi:10.1016/j.jece.2023.111741.

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