



Full Length Article

Photocatalytic hydrogen production and photodegradation of organic dyes of hydrogenated TiO₂ nanofibers decorated metal nanoparticlesMing-Chung Wu^{a,*,b,c}, Wei-Kang Huang^a, Ting-Han Lin^a, Yu-Jen Lu^{d,e,*}^a Department of Chemical and Materials Engineering, Chang Gung University, Taoyuan 33302, Taiwan^b Green Technology Research Center, Chang Gung University, Taoyuan 33302, Taiwan^c Division of Neonatology, Department of Pediatrics, Chang Gung Memorial Hospital, Linkou, Taoyuan 33305, Taiwan^d Department of Neurosurgery, Chang Gung Memorial Hospital, Linkou, Taoyuan 33305, Taiwan^e College of Medicine, Chang Gung University, Taoyuan 33302, Taiwan

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ABSTRACT

A series of metal decorated hydrogenated TiO₂ nanofibers (metal-H:TiO₂ NFs) are synthesized successfully in this study. It is synthesized by the hydrothermal approach and combined with wet impregnation method in thermal hydrogen reduction. Nine types of metal nanoparticles, Ag, Co, Cr, Cu, Fe, Ni, Pd, Pt and Y, were used to decorate on the surface of H:TiO₂ to enhance the photocatalytic activity. The photocatalytic performance of various metal-H:TiO₂ NFs was estimated by the photodegradation of organic dyes. Ag-H:TiO₂ NFs gives the fastest decoloration rate of brilliant green among others. Moreover, the evolution rate of H₂ over Pd-H:TiO₂ NFs under UV-A and UV-B illumination is 17,000 μmol/g-h and 25,600 μmol/g-h, respectively, which corresponding to photoefficiency values of ~7.54% and ~11.35%. The metal-H:TiO₂ NFs developed in this study can be a facile and environmentally friendly way for searching the high-performance photocatalysts in the field of environmental and energy issue.

1. Introduction

Titanium dioxide (TiO₂) has attracted broad research attention because of their applications in environmental pollution removal and photocatalytic hydrogen generation in the past few decades [1–13]. The one-dimensional titanate nanomaterials synthesized from TiO₂ powders in alkaline solutions has developed a new trend for the synthesis of TiO₂ for large-scale TiO₂-based nanofibers [14–20]. Nanofibers is much easier to achieve a percolated electrical network with elongated materials than regular nanoparticles. Furthermore, a bunch of nanofibers leads to the mechanical strength in tangled networks and shows the heavier entanglement with other nanoparticles [11,18,21–27].

Hydrogenated TiO₂ (H:TiO₂) exhibited strong visible light absorbance, a disordered surface enriched in trap states, and remarkably improved photocatalytic performance. The hydrogenation process for TiO₂ is usually obtained by annealing in a tube furnace filled with high-pressure hydrogen accompanied by a long calcination time. However, the progress in high-pressure and high-temperature conditions could be a danger for large-scale production [28,29]. Since hydrogenation process was used to fabricate hydrogenated TiO₂, many scientists used this method to modify the bandgap to expand absorption behavior, and

H:TiO₂ has activated a lot of interest in photocatalytic hydrogen production [30–32]. Hydrogenation was demonstrated as a direct and effect approach to obtain high-performance H:TiO₂ nanotubes and nanowires for photocatalytic hydrogen evolution [33,34]. Moreover, through introducing surface defects (oxygen vacancy and Ti³⁺ interstitial defect) on the TiO₂ surface by hydrogenation process, it will result in the highly localized midgap states which could enhance the photocatalytic performance by the appropriate separation of electron and hole in black TiO₂ [35,36]. Many research groups also reported that H:TiO₂ has been applied in visible-light photocatalytic application, such as the photo-oxidation of organic compound and photocatalytic hydrogen production [37–40].

Noble metal decoration on TiO₂ surface is a convenient method to enhance the photocatalytic activity. The noble metal can enhance the light-induced exciton separation efficiently due to the large work function of noble metal including silver (Ag), gold (Au), iridium (Ir), palladium (Pd), platinum (Pt), rhodium (Rh), and ruthenium (Ru), which can form Schottky barrier between TiO₂ and metal nanoparticles [41–49]. For certain metal nanoparticles can induce localized surface plasmon resonance. It increases the absorption of TiO₂ at certain wavelengths depend on the metal types [50–57]. Specific metals and metal

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oxides decorated on TiO₂ surface is an efficient approach to synthesize high activity TiO₂-based photocatalysts.

As photocatalytic science and technology has been widely developed in recent decades, doping noble metals on TiO₂ photocatalysts has been proved that specific interface between noble metals and TiO₂ can enhance the activity. Nevertheless, the H:TiO₂ photocatalytic property based on different metal NPs has not been revealed and discussed thoroughly. Which metal decorated on H:TiO₂ NFs surface perform the high photocatalytic performance on photodegradation of organic dye and photocatalytic hydrogen production is still unknown at this stage. Herein, in order to combine the merits of hydrogenated TiO₂ and noble metal decoration and enhance the photocatalytic performance, we explored the application of H:TiO₂ by integrating metal NPs on its surface.

In this study, various metal decorated hydrogenated TiO₂ nanofibers (metal-H:TiO₂ NFs) were synthesized by combining the hydrothermal process and wet-impregnation method, followed by one-step thermal treatment in low H₂ partial pressure. The optimal calcination condition is achieved to enhance the photocatalytic performance. A series of metal-H:TiO₂ NFs were applied for degradation of hazardous organic dye and photocatalytic hydrogen evolution under UV irradiation and discussed in detail. The developed metal-H:TiO₂ NFs could be a potential material in the fields of environmental pollutants removal and photocatalytic hydrogen production. Here, we report on the synthesis of H:TiO₂ NFs and their metal (Ag, Co, Cr, Cu, Fe, Ni, Pd, Pt, and Y)-decorated derivatives. To the best of our knowledge, Pd-H:TiO₂ NFs presents an outstanding H₂ evolution performance in water-ethanol solutions under UV-A and UV-B illumination [45].

2. Experimental details

The hydrogen sodium titanate nanofibers (H_xNa_{2-x}Ti₃ NFs) is prepared according to our previous study [58]. Nine types of metal precursors were used to decorate on the surface of H_xNa_{2-x}Ti₃O₇ NFs. For the 1.0 wt% metal decoration, 9.5 mg of silver nitrate (AgNO₃, CHONEYE, extra pure reagent), 30.2 mg of cobalt (II) nitrate hexahydrate (Co(NO₃)₂·6H₂O, ACROS, 99%), 44.5 mg of chromium(III) nitrate nonahydrate (Cr(NO₃)₃·9H₂O, ACROS, 99%), 25.5 mg of copper(II) acetylacetonate (Cu(C₅H₇O₂)₂, ACROS, 98%), 44.7 mg of iron(III) nitrate nonahydrate (Fe(NO₃)₃·9H₂O, ACROS, 98 + %), 30.3 mg of nickel (II) nitrate hexahydrate (Ni(NO₃)₂·6H₂O, ACROS, 99%), 12.8 mg of palladium (II) acetate (Pd(OCOCH₃)₂, ACROS, 99.9%), 12.5 mg of platinum (II) acetylacetonate (Pt(C₅H₇O₂)₂, ACROS, 98%) and 26.1 mg of yttrium(III) nitrate hexahydrate (Y(NO₃)₃, ACROS, 99.9%) were firstly dissolved in 200 mL of ethanol/acetone (1:1, v/v), separately. Then, 594 mg of H_xNa_{2-x}Ti₃O₇ NFs was suspended in the precursor solution using 3 h of ultrasonic agitation. The mixture of various metal precursors and H_xNa_{2-x}Ti₃O₇ NFs were stirred for 6 h, respectively. After evaporating the solvent at ~80 °C under N₂ atmosphere, the samples were undergo a calcination treatment in 15% H₂ (in N₂ buffer) flow for 12 h to obtain the various metal decorated hydrogenated TiO₂ nanofibers (metal-H:TiO₂ NFs). The illustration of the preparation of pristine TiO₂ NFs, H:TiO₂ NFs and various metal-H:TiO₂ NFs is shown in Fig. 1. The metal loading is ~1.0 wt% for each metal-H:TiO₂ NFs.

Absorption spectrophotometer (JASCO, V-650, Japan) is used to acquire the UV-Vis reflectance spectra of synthesized materials. X-ray diffractometer (Bruker, D2 phaser with Xflash 430) was used to determine the crystal structures of the synthesized materials using Cu K_α radiation at 50.0 kV and 300.0 mA. XRD patterns were collected from 2θ between 10 and 80 with a 0.005° step at 5° min⁻¹. For Raman scattering spectra of the TiO₂-based materials and the commercial anatase TiO₂ powder (Acros, 98 + %, anatase powder), they were positioned on a high-resolution piezoelectric stage of the scanning microscopy (WiTec, Alpha 300S) and were excited by a 632.8 nm He-Ne laser (25 mW). The laser beam was focused with a 10× objective lens (Nikon plane objective, NA ≈ 0.9), and the focused laser beam size was about 10 μm. Spherical-aberration corrected field-emission

transmission electron microscope (JEOL, JEM-ARM200FTH, Japan) is adopted to investigate the microstructure of the synthesized materials.

Photodegradation activity was determined by evaluating the decolorization of brilliant green. First of all, 20.0 mg TiO₂ catalyst was dispersed in 150 mL of brilliant green solution (10.0 mg/L). After 30 min adsorption process in the dark to eliminate the deviation, the mixture solution was irradiated under two lamps of UV-A light and vigorously stirred at the same time. These specimens were centrifuged at 5000 rpm for 15 min to remove the remaining TiO₂ catalyst. The concentration of brilliant green in the supernatant was quantified at the absorbance of λ = 624 nm by UV-Vis spectrophotometer (Jasco Analytical Instruments, V-650 UV-Vis Spectrophotometer) and further calculated based on Beer-Lambert law. On the other hand, photocatalytic hydrogen production was conducted in the home-made system under UV-B and UV-A irradiation.[59] 10.0 mg TiO₂ catalyst was dispersed in 2.0 L mixture of ethanol and water which volumetric ratio is 1:1. For the photostability of the TiO₂ catalysts, the photocatalytic hydrogen production was performed in five cycles via the same condition and parameter. To study the photoelectrochemical properties, the linear sweep voltammetry (LSV), transient photocurrent, and electrical impedance spectroscopy (EIS) were analyzed by a potentiostat (Metrohm, Autolab PGSTAT302N) with a standard three-electrode system including Ag/AgCl in KCl as a reference electrode, Pt counter electrode, and TiO₂-coated working electrode. 0.50 M sodium sulfate aqueous solution was used as electrolyte. The 1.0 cm² TiO₂ electrode was fabricated via blade coating method to lie the paste on the FTO, which paste consisted of 0.10 g TiO₂ catalyst, 2.50 mL ethanol, 0.50 mL water and few Triton X-100 (Sigma-Aldrich). The pristine TiO₂ electrode was calcined at 350 °C for 3 h in the air, while H:TiO₂ and metal-H:TiO₂ electrode were calcined at 350 °C for 3 h in the air followed by thermal reduction in the 15% H₂ (in N₂ buffer). For the LSV measurement, applied potential ranged from -0.5 to 1.5 V with scan rate of 50 mV·s⁻¹. The transient photocurrent was detected with a potential of 0 V. In the EIS, the potential was set as 0 V with an amplitude of 0.01 V, and frequency ranged from 1 to 10⁴ Hz.

3. Results and discussion

The various TiO₂-based nanofibers calcined at different combinations of temperature and time are prepared to find the optimal calcination process for high-performance photocatalyst. The TiO₂ nanofibers calcined in air and in the flow of H₂/N₂ at various temperatures for 12 h are denoted as TiO₂-X NFs and H:TiO₂-X NFs respectively, for which “X” indicating the calcination temperature. Photos of pristine TiO₂ NFs and H:TiO₂ NFs calcined from 450 °C to 700 °C are shown in Fig. 2a and b. The pristine TiO₂ NFs calcined in air shows white color, but H:TiO₂ NFs calcined in the flow of H₂/N₂ is black. H:TiO₂-NFs and pristine TiO₂ NFs calcined below 450 °C could show the existence of carbon. Therefore, TiO₂ nanocrystals calcined above 450 °C are adopted for further study. From Fig. 2c and d, we can observe that the reflectance spectra clearly enhance in the visible region, when H:TiO₂ NFs is calcined in the flow of H₂/N₂ mixture. For H:TiO₂ NFs, the reflectance spectra reveal an enhancement compared with pristine TiO₂ NFs due to the oxygen atom defects and/or trivalent titanium formation [60].

The phase transformation and vibrational modes of the pristine TiO₂ NFs and H:TiO₂ NFs with various calcination temperatures are investigated by Raman scattering spectroscopy. From Fig. 3, Raman scattering spectra show that H:TiO₂ NFs transform to pure TiO₂ anatase phase when calcination temperature of 650 °C is reached. Raman spectra of anatase TiO₂ powder (Acros, 98 + %) was measured as the standard for structural identification. Anatase TiO₂ powder shows the six Raman-active modes, including E_g (144 cm⁻¹), E_g (200 cm⁻¹), B_{1g} (398 cm⁻¹), A_{1g} (515 cm⁻¹), B_{1g} (517 cm⁻¹), and E_g (640 cm⁻¹), of anatase phase, respectively [61]. For TiO₂-based material, there is an increased crystallinity as calcination temperature is raised. At the same calcination temperature, the signal of E_g band at 144 cm⁻¹ of H:TiO₂

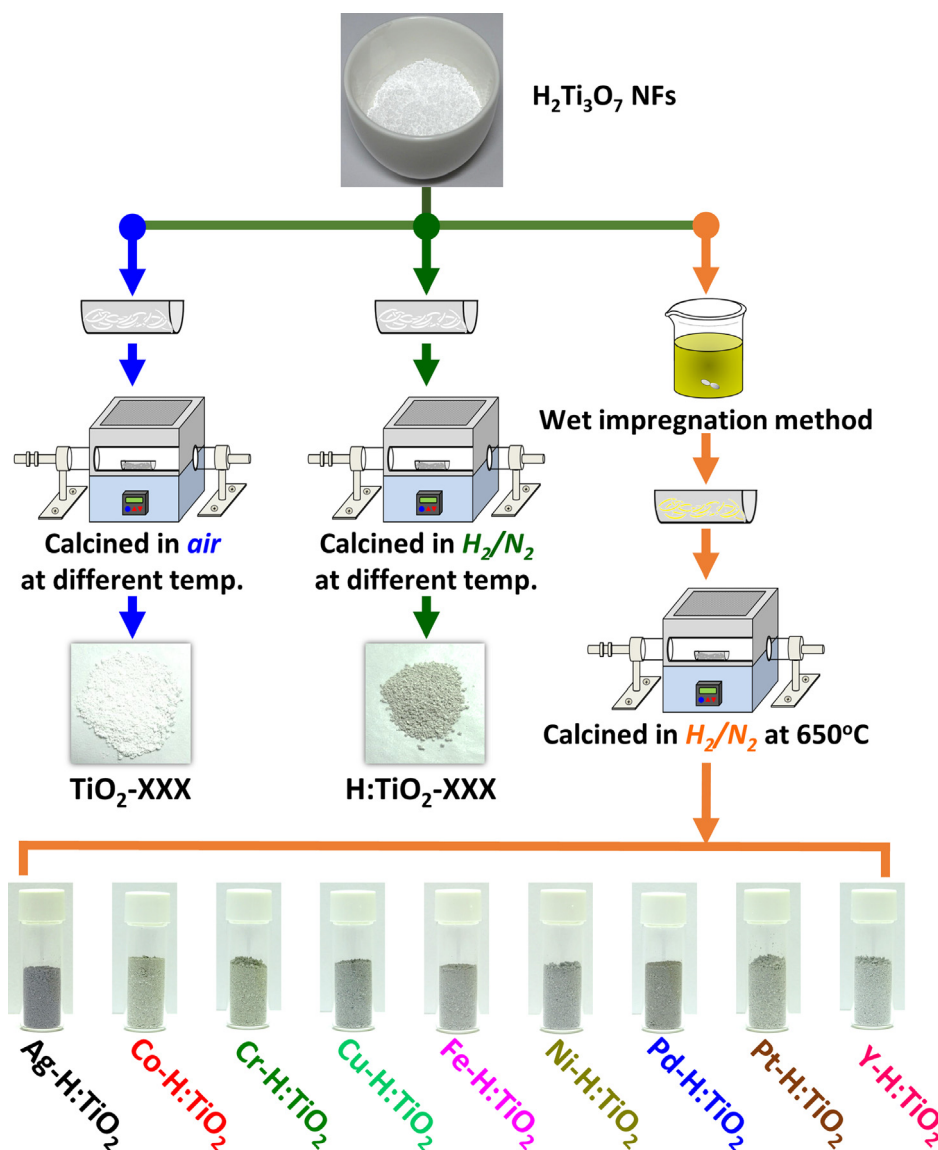


Fig. 1. Illustration of the preparation of pristine TiO_2 NFs, H:TiO_2 NFs and various metal- H:TiO_2 NFs.

NFs is weaker than pristine TiO_2 NFs, because the titanium trivalent ion and oxygen vacancies formation could be introduced through the calcination process in H_2/N_2 [28,60].

The crystal structure analysis for pristine TiO_2 NFs series and H:TiO_2 NFs series are characterized using X-ray diffractometer. The XRD patterns of TiO_2 NFs under the calcination procedure in the flow of H_2/N_2 are shown in Fig. 4. Below the calcination temperature of 650 °C, the intensity of (1 0 1) reflection at 2θ of 25.3° for H:TiO_2 NFs is increased with ascending the calcination temperature which is due to the high-temperature calcination process could improve the lattice ordering of anatase TiO_2 . However, pristine TiO_2 NFs and H:TiO_2 NFs would transform to rutile phase TiO_2 at the calcination temperature of 700 °C. The intensity of (1 0 1) reflection at 2θ of 25.3° is thus decreased. For H:TiO_2 NFs series, the sample with the calcination temperature of 650 °C ($\text{H:TiO}_2\text{-650}$ NFs) exhibits a high degree of crystallinity with the crystalline phase anatase TiO_2 . The $\text{H:TiO}_2\text{-650}$ NFs pattern could be indexed as the body-centered tetragonal unit cell of anatase TiO_2 [COD ID:720675]. The XRD study is in accordance with the Raman measurement. Hence, we adopted the calcination temperature of 650 °C to prepare the samples for further study.

The enhancement of photocatalytic activity gained through the hydrogenated procedure is revealed by photodegradation test of

brilliant green under UV-A irradiation. To analyze the photodegradation performance, the retain concentration of brilliant green is calculated according to a calibration curve measured at $\lambda = 625$ nm in the absorption spectra. The photodegradation of brilliant green over H:TiO_2 follows first-order kinetics which is simplified from Langmuir-Hinshelwood kinetics at lower initial organic dye concentration. The kinetic constant, k , can be defined as $\ln\left(\frac{C}{C_0}\right) = -kt$; where C_0 and C represent the concentration of brilliant green at initial and time t , respectively, and k is the apparent reaction rate constant [4,62]. The photodegradation of brilliant green ($\frac{C}{C_0}$ curves) at various times over pristine $\text{TiO}_2\text{-650}$ NFs and various H:TiO_2 NFs and are shown in Fig. 5. Linearized kinetic plots of $\ln\left(\frac{C}{C_0}\right)$ versus t for photodegradation of brilliant green indicates the results are in good agreement with the Langmuir-Hinshelwood model (Fig. 5b). The linear slopes for each H:TiO_2 represent k , the reaction rate constant, which directly reveals the performance of photodegradation. Among pristine TiO_2 NFs series, pristine $\text{TiO}_2\text{-650}$ NFs shows the fastest decolorization rate for the photodegradation test, and its calculated reaction rate constant is $\sim 0.097 \text{ min}^{-1}$. Interestingly, $\text{H:TiO}_2\text{-650}$ NFs exists the highest calculated reaction rate constant of $\sim 0.111 \text{ min}^{-1}$ (Fig. 5c).

We demonstrate HRTEM study to reveal why H:TiO_2 shows high

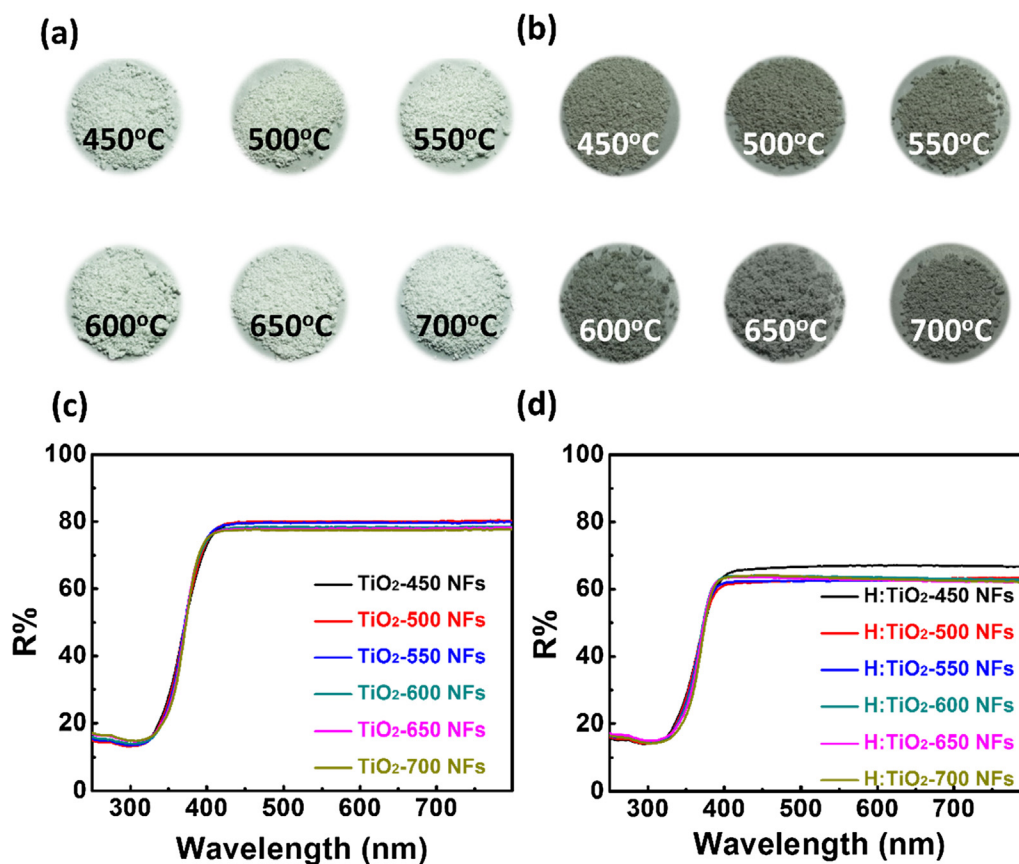


Fig. 2. (a, b) Photographs and (c, d) reflectance spectra of pristine TiO₂ NFs series (a, c) and H:TiO₂ NFs series (b, d) calcined at various temperatures in the flow of H₂/N₂ for 12 h.

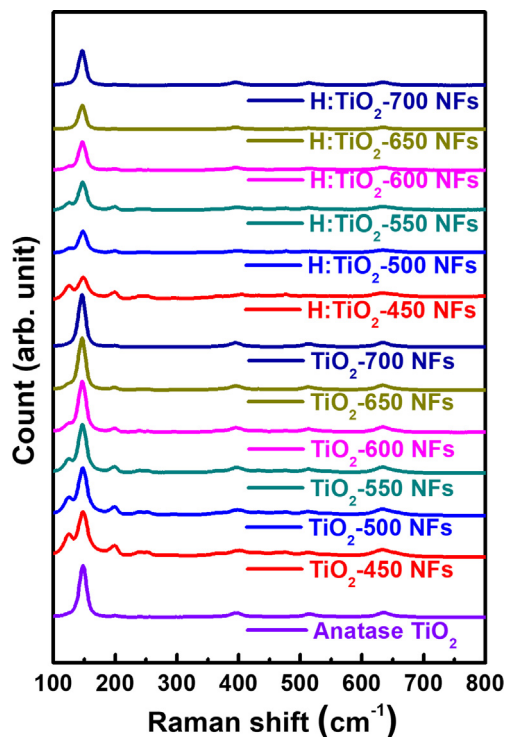


Fig. 3. Raman scattering spectra of pristine TiO₂ NFs series and H:TiO₂ NFs series calcined at various temperatures for 12 h and the commercial anatase TiO₂ powder.

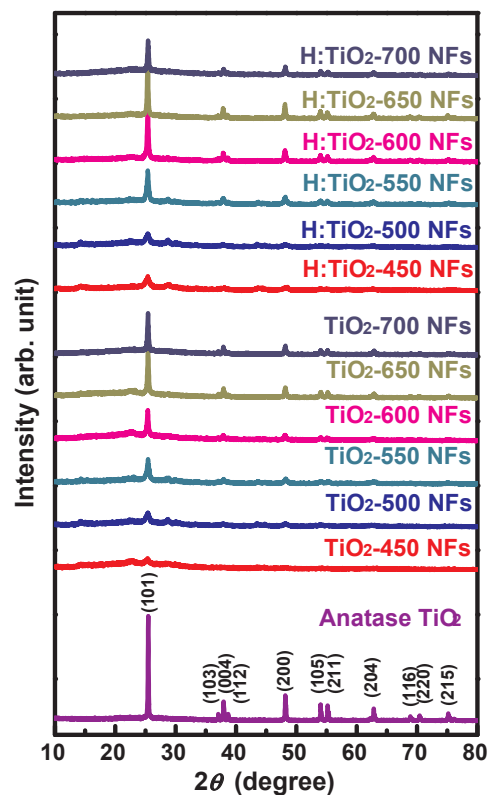


Fig. 4. XRD patterns of pristine TiO₂ NFs series and H:TiO₂ NFs series calcined at various temperatures for 12 h and the commercial anatase TiO₂ powder.

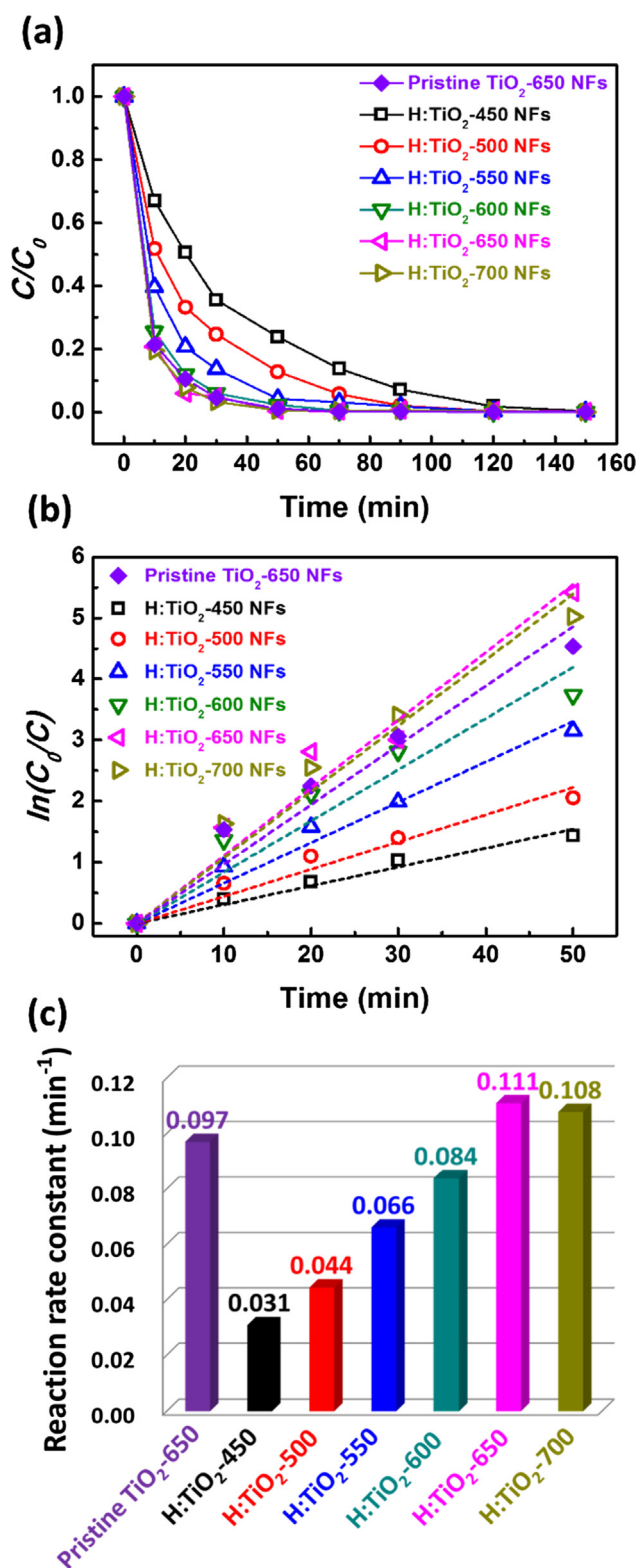
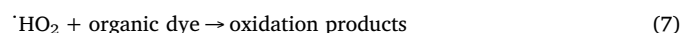
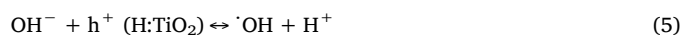
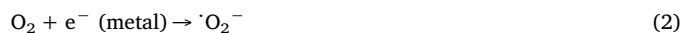
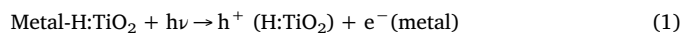


Fig. 5. (a) The C/C_0 curves, (b) the linearized kinetic plots and (c) the reaction rate constants for the photodegradation of brilliant green using pristine TiO₂-650 and H:TiO₂ NFs series under UV-A irradiation.

performance in the photodegradation of brilliant green compared with pristine TiO₂-650 NFs (Fig. 6). The high-magnification lattice images are shown in Fig. 6b and d for pristine TiO₂-650 NFs and H:TiO₂-650 NFs, respectively. The insets of Fig. 6b and d shows the fast Fourier transformed patterns of both specimens for observing the direction of

(1 0 1) crystal plane. The d_{101} of pristine TiO₂-650 NFs (3.50 Å) is larger than d_{101} of H:TiO₂-650 NFs (3.45 Å). The statements suggest that calcination in the flow of H₂/N₂ resulted in the oxygen vacancy, trivalent titanium (Ti³⁺) and/or structural defect in the TiO₂ lattice. According to the literature, introducing the surface defect mentioned above on the surface of TiO₂ could impart the suppression of charge carrier recombination and enhanced optical absorption. Therefore, the excellent control of the microstructure can still warrant additional research effort from a materials science point of view.

Adding a proper co-catalyst is a facile method to activate the photodegradation performance and photocatalytic hydrogen production. Here, nine types of metal nanoparticles, Ag, Co, Cr, Cu, Fe, Ni, Pd, Pt, and Y, with a concentration of ~1.0 wt% are used to decorate on the surface of H:TiO₂-650 NFs. After the decoration procedure, the interface between the metal nanoparticle and TiO₂ would form a Schottky barrier which could facilitate the charge carrier transport. According to the mechanism of the photodegradation of organic dye, the more electron transfers to the surface could generate adequate reactive oxygen species (ROS) which seen as major factors for the photodegradation reaction [3]. The photodegradation test of brilliant green under UV-A irradiation is demonstrated here to reveal which of the metal nanoparticle could be a suitable candidate for enhancing the photocatalytic activity (Fig. 7). According to linearized kinetic plots fitted form $\left(\frac{C}{C_0}\right)$ curves (Fig. 7a and b), the Ag-H:TiO₂ NFs gives the fastest decolorization phenomenon. For the H:TiO₂ decorated with Ag, Pd, Pt and Y, they exhibit the faster decolorization phenomenon compared with nonmetal decorated H:TiO₂-650 NFs (~0.111 min⁻¹) over the photodegradation of brilliant green (Fig. 7c). The other metal decorated H:TiO₂-650 do not have suitable electronic properties for useful electron transfer reactions. The metal nanoparticles which negatively charged could lead to chemical reduction of protons on the surface; on the other hand, the metal nanoparticles which positively charged could enhance the oxidative process. Herein, we proposed a possible photodegradation pathway shown as below [60,63].



As the metal-H:TiO₂ was irradiated by the light with sufficient energy, the electron-hole pairs will form and separate. The electrons will prefer to transfer through the defect area then to schottky barrier and finally into metal NPs; the holes will stay in the bulk TiO₂. Subsequently, a series of redox reaction occur and produce the several ROS such as $\cdot\text{O}_2^-$, $\cdot\text{HO}_2$, OH^- , and $\cdot\text{OH}$. Finally, these radicals will oxidize the organic dye and convert it to harmless products.

The photocatalytic hydrogen evolution of various metal-H:TiO₂ NFs are investigated under the UV-A and UV-B light irradiation, respectively. In Fig. 8a, both of Pd- and Pt-decorated on H:TiO₂ NFs present the excellent hydrogen production rates, which are ~9000 and 17,000 $\mu\text{mol/g}\cdot\text{h}$ under UV-A illumination. It means these two kinds of noble metal decoration are beneficial for achieving the high photocatalytic activity as illumination. Furthermore, the photocatalytic hydrogen production of as-prepared TiO₂ NFs irradiated by UV-B light are measured and the corresponding evolution rates are demonstrated in Fig. 8b. Interestingly, Ni-H:TiO₂ NFs also shows the ability to produce the hydrogen and its evolution rate is seen to be about 1900 $\mu\text{mol/g}\cdot\text{h}$. Pt-H:TiO₂ NFs remains the similarly high rate of 14,900 $\mu\text{mol/g}\cdot\text{h}$ under UV-B illumination. Meanwhile, Pd-H:TiO₂ NFs exhibits the highest

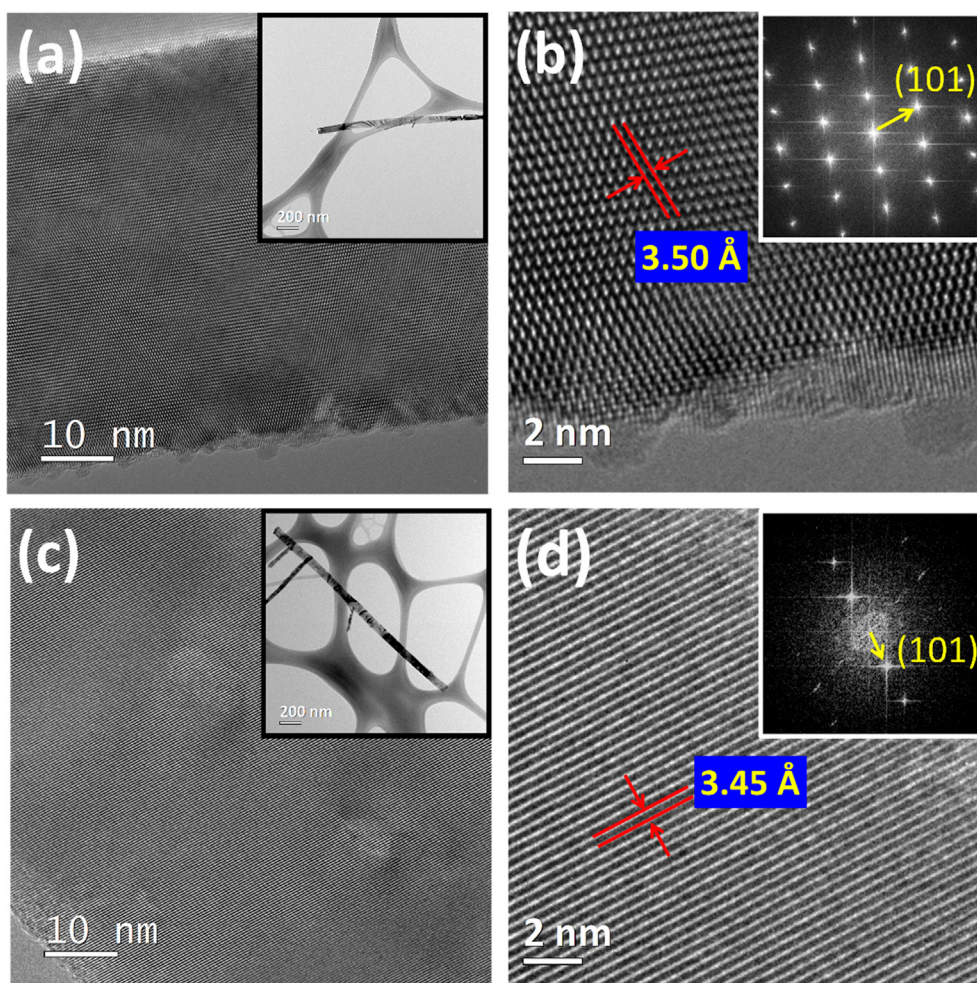


Fig. 6. HRTEM images of (a, b) pristine TiO_2 -650 NFs and (c, d) H:TiO_2 -650 NFs, and the insets of (b, d) are the corresponding fast Fourier transformed pattern of pristine TiO_2 -650 NFs and H:TiO_2 -650 NFs.

production rate at $25,600 \mu\text{mol/g}\cdot\text{h}$. After the five cycles, it still maintains $\sim 72\%$ of hydrogen evolution rate as shown in Fig. S1. In this case, the photocatalytic H_2 evolution over most metal- H:TiO_2 showed suppressed and low activities compared to photodegradation performance. As we know, photocatalytic H_2 evolution is a typical uphill reaction which needed additional energy to activate the reaction [64,65]. A suitable band alignment toward the water splitting and abundant electron/hole pairs forming become important issues. However, the H_2 evolution over all metal- H:TiO_2 NFs under visible light illumination is too low to be detected. The low energy and intensity of visible light cannot entirely excited the metal- H:TiO_2 NFs to further produce sufficient amounts of electron/hole pairs. To evaluate the photoenergy conversion efficiency, which can be identified that how much photoenergy converted to chemical energy stored in hydrogen for UV-A and UV-B photons, the general definition is described as following,

$$\text{Photo energy conversion efficiency (\%)} = \frac{\text{output energy of hydrogen evolved}}{\text{energy of incident light}} \quad (8)$$

The photoenergy conversion efficiency of Pd- H:TiO_2 NFs under UV-A irradiation and under UV-B irradiation are $\sim 7.54\%$ and $\sim 11.35\%$, respectively. Herein, the brief conclusion according to as-presented results is that Pd- H:TiO_2 NFs and Pt- H:TiO_2 NFs under UV irradiation perform a strong ability to produce the hydrogen. It can be attributed that the same series of Pd and Pt at group 10 element of periodic table always shows the similar electron configuration and chemical

properties such as suitable work function for TiO_2 heterojunction [4]. The theoretical work function of as-mentioned metals reported by previous literature were summarized in Table S1. As these metals are decorated on the surface of TiO_2 , the Schottky barrier is formed on the interface. Typically, the barrier heights for TiO_2 NFs ($\chi = 3.90 \text{ eV}$) decorated with palladium ($\phi_{\text{Pd}} = 5.60 \text{ eV}$), platinum ($\phi_{\text{Pt}} = 5.70 \text{ eV}$), and nickel ($\phi_{\text{Ni}} = 5.15 \text{ eV}$) [66] might be 1.70 eV, 1.80 eV, and 1.25 eV, respectively. It is effective to promote the water splitting reaction and rectify the electron-hole recombination [4]. In addition, combining the advantages of surface defects on H:TiO_2 NFs and the metal-induced Schottky barrier, the flows of electron-hole pairs might be rectified and the electrons effectively migrate to metal. Therefore, both of Pd- H:TiO_2 and Pt- H:TiO_2 NFs reveal high hydrogen evolution rate.

To further investigate the charge dynamic in the metal- H:TiO_2 , photoelectrochemical properties were studied by several ways including photo-assisted LSV, EIS and transient photocurrent measurement. Fig. 9 presents the EIS Nyquist plot of three kinds of TiO_2 -based photocatalyst. Typically, the arc radius of EIS depicts the resistance of the interface layer on the electrode.[67,68] The small arc radius of all TiO_2 photocatalysts were found under irradiation, indicating that the lower resistance is due to the more production of charge carrier compared to the condition in the dark. Fig. 10a shows the photocurrent response in the dark and under UV irradiation over pristine TiO_2 , H:TiO_2 , and Pd- H:TiO_2 . Under the UV irradiation, the photocurrent density of all TiO_2 electrodes exhibit a significant improvement, and Pd- H:TiO_2 presents the highest response. It is attributed to the metal decoration which could further enhance the light harvesting as well as

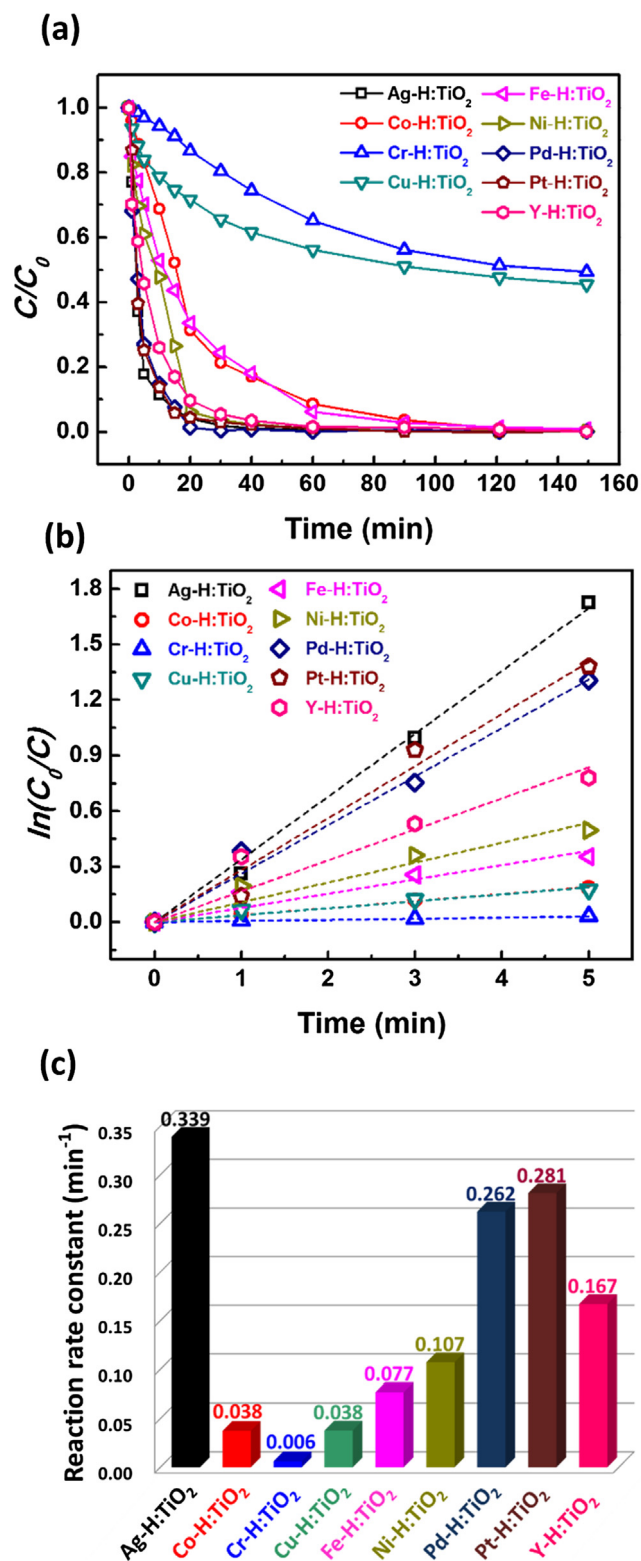


Fig. 7. (a) The C/C_0 curves, (b) the linearized kinetic plots and (c) the reaction rate constants for the photodegradation of brilliant green using various metal-H:TiO₂ NFs.

transport the charge efficiently.[67,68] According to Fig. 10b of transient photocurrent, the current density not only increased dramatically but also addressed the delay in the circuit when the transient photocurrent generated. Interestingly, H:TiO₂ promotes the electrons transfer to and retain on the surface defect (oxygen vacancy and Ti³⁺ interstitial

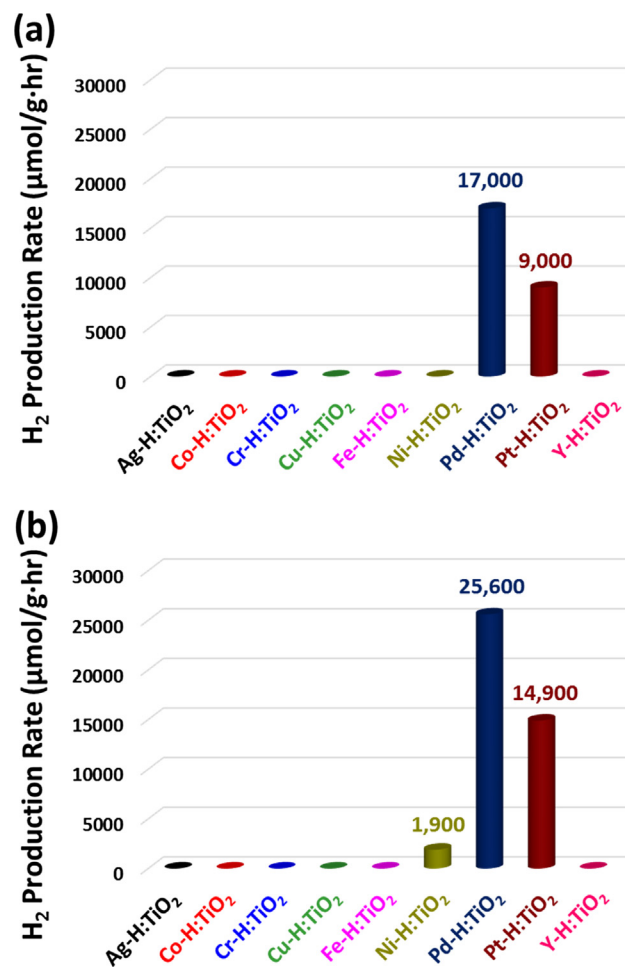


Fig. 8. The hydrogen evolution rates of various metal-H:TiO₂ NFs under the following light source illumination: (a) UV-A, and (b) UV-B.

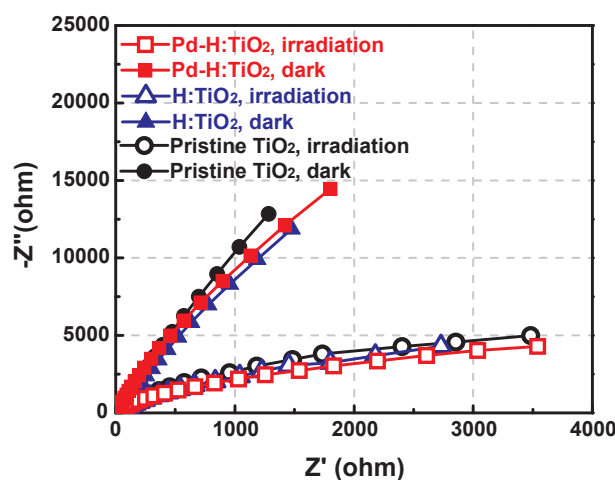


Fig. 9. EIS Nyquist plot of various TiO₂ in the dark and under the UV irradiation.

defect) [60], which could further delay the photocurrent transition. Therefore, we believe that metal decoration is quite beneficial to charge transport and light harvesting, and demonstrate the superiority of Schottky barrier in the metal-H:TiO₂. The chemical state analysis of various TiO₂ catalyst analyzed by X-ray photoelectron spectroscopy (Fig. S2) suggested the more intense Ti³⁺ signal located around 457.8 eV in the H:TiO₂ and metal-H:TiO₂ owing to the Ti³⁺ interstitial

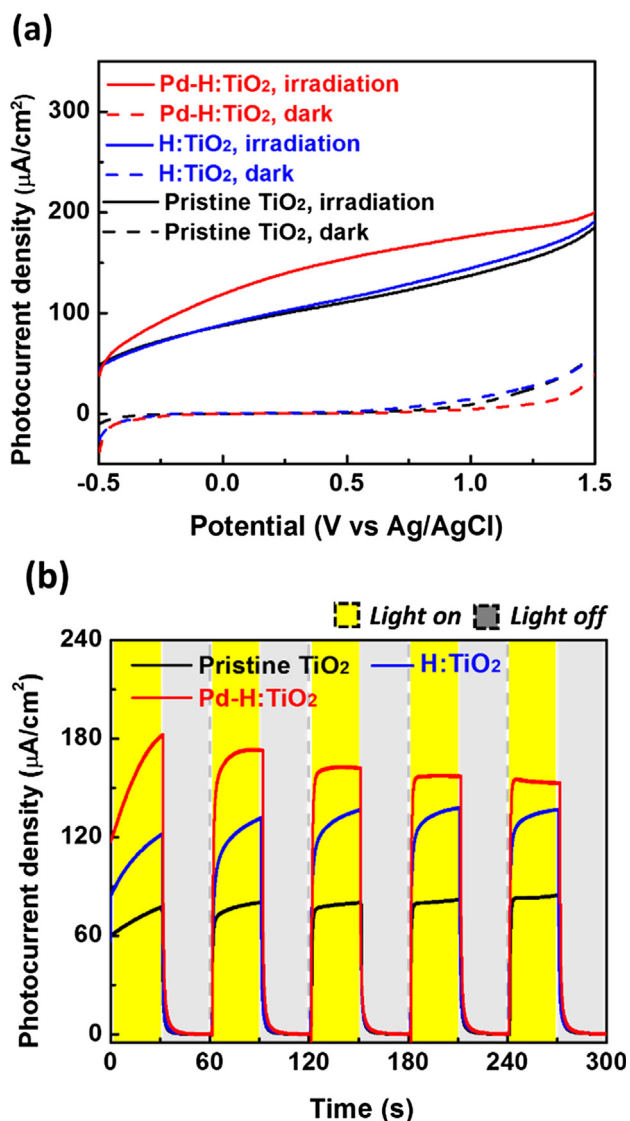


Fig. 10. (a) Photocurrent response measured by LSV of the pristine TiO₂, H:TiO₂, and Pd-H:TiO₂ in the dark and under UV-B irradiation. (b) Transient photocurrent of pristine TiO₂, H:TiO₂, and Pd-H:TiO₂ under UV-B irradiation.

defects.[60] In addition, the composition of palladium were about 0.49 at%. The BET specific surface area and the pore distribution were summarized in Fig. S3 and Table S2. Among three kinds TiO₂ photocatalyst, H:TiO₂ presents the largest surface area which is $\sim 34.21 \text{ m}^2 \text{ g}^{-1}$ because of much more surface defects. The surface area of Pd-H:TiO₂ remained $31.80 \text{ m}^2 \text{ g}^{-1}$ suggested that these surface defects were overlaid by the metal nanoparticles which is about 5.0 nm, characterized by HRTEM as shown in Fig. S4.

The mechanism of photocatalytic hydrogen evolution from ethanol aqueous solution over various metal-H:TiO₂ NFs are illustrated in Fig. 11. Once the metal-H:TiO₂ NFs is illuminated by light, the photocatalytic reaction initiates. Photon with the sufficient energy intensity is capable to generate the photoexcited electron-hole pair. During the carrier migration processes, some of the photoexcited charges may recombine. To avoid the rapid recombination of electron-hole pair, the introduced surface defects provide trapping sites for photoexcited carriers. In the meanwhile, the separated hole would react with ethanol which acts as sacrificial agents to form proton which would further react with electron to generate H₂. In addition, the metal nanoparticles with the appropriate chemical states decorated on the surface of H:TiO₂ NFs acts as electron reservoirs to capture the photogenerated electron on TiO₂ and the electron subsequently react with proton to form hydrogen. With the cooperation of Schottky interface between TiO₂ and the metal nanoparticle, the large work function of metal nanoparticles could facilitate the electron that trapped by oxygen vacancy and Ti³⁺ interstitial defect to transport to the surface of metal nanoparticles. Furthermore, the existence of Schottky barrier could prevent the electron restricted in metal nanoparticle inject back to semiconductor. Therefore, we can expect that the metal-H:TiO₂ NFs possesses a more efficient charge separation and a lower electron-hole recombination rate [69].

4. Conclusion

Many types of metal nanoparticles were chosen to decorate on H:TiO₂ NFs surface to enhance the photodegradation activity and the photocatalytic hydrogen production rates. For each metal-H:TiO₂ NFs, the metal loading is $\sim 1.0 \text{ wt}\%$. For the photodegradation of organic dye, Ag-H:TiO₂ NFs shows the fastest decoloration phenomenon. For the photocatalytic hydrogen production, the highest hydrogen production rate was produced by Pd-H:TiO₂ NFs at a rate of $\sim 17,000 \mu\text{mol}/\text{g}\cdot\text{h}$ under UV-A irradiation and at $\sim 25,600 \mu\text{mol}/\text{g}\cdot\text{h}$ under UV-B irradiation corresponding to the photo energy conversion efficiency of $\sim 7.54\%$ and $\sim 11.35\%$, respectively. Ag-H:TiO₂ NFs and

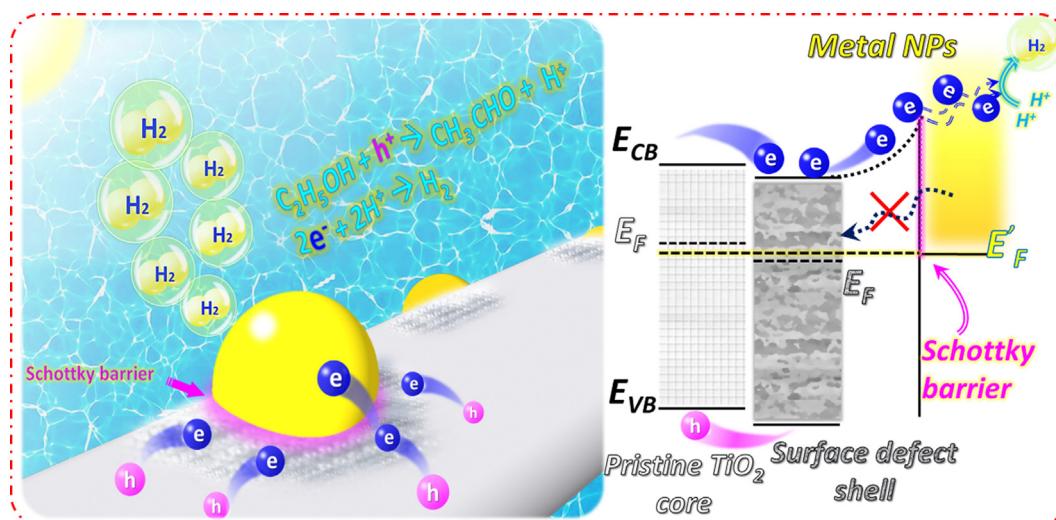


Fig. 11. The mechanism of photocatalytic hydrogen evolution from ethanol-water mixture over various metal-H:TiO₂ NFs.

Pd-H:TiO₂ NFs synthesized in our study represent the typical metal-semiconductor heterostructure systems that could be widely developed in this study can be a facile and environmentally friendly way for searching the high-performance photocatalysts in the field of environmental and energy issue.

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Appendix A. Supplementary material

Supplementary data to this article can be found online at <https://doi.org/10.1016/j.apsusc.2018.10.240>.

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